# Density of the System $KF + K_2NbF_7 + Nb_2O_5$

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The density of melts of the system  $KF(1)+K_2NbF_7(2)+Nb_2O_5(3)$  up to 20 mol %  $Nb_2O_5$  has been measured using the Archimedean method. On the basis of the measured density values, the molar volumes of the melts and partial molar volumes of  $Nb_2O_5$  in different binaries were calculated. The density in the system  $KF(1)+K_2NbF_7(2)+Nb_2O_5(3)$  increases from KF through  $K_2NbF_7$  to  $Nb_2O_5$ . In all three binary systems, some interactions have been determined.

### Introduction

The molten system  $KF(1) + K_2NbF_7(2) + Nb_2O_5(3)$  is interesting from both the theoretical and technological points of view. The melts of this system are promising electrolytes for the electrochemical deposition of smooth and adherent niobium layers on a metallic base.<sup>1</sup>

A crucial problem in niobium deposition in molten salts is the presence of oxygen in the electrolyte because it is extremely difficult to prepare a melt free of  $O^{2-}$  ions, especially for industrial applications. Formerly, it was assumed that the presence of  $O^{2-}$  ions may lower the quality of Nb coatings or even prevent the formation of Nb coatings. The influence of  $O^{2-}$  ions on the reduction mechanism of Nb and the formation of niobium–oxofluoride complexes in the melt has been studied elsewhere.<sup>2-6</sup>

Though electrolyte impurities, such as oxides, hydroxides, chlorides, bromides and iodides, were formerly considered to be undesirable,<sup>7</sup> later it was found that the presence of small amounts of oxides in the melt increases the current efficiency during electrolysis. Christensen et al.<sup>4</sup> obtained the highest current efficiency in melts with O/Nb molar ratios in the range of  $1 < n_0/n_{\rm Nb(V)} < 0.5$ . The presence of oxide in the melt causes the formation of various oxofluoro complexes of niobium. It was shown that relatively pure Nb coatings can be obtained from FLINAK melts if  $n_0/n_{\rm Nb(V)}$  is less than 1. However, even a small number of O<sup>2-</sup> ions can entirely change the mechanism of Nb deposition<sup>2,3</sup> depending on the types of niobium– oxofluorides formed in the melt.

The phase diagram<sup>9-11</sup> and the density<sup>12</sup> of the system  $KF(1) + K_2NbF_7(2)$  have been studied. The congruently melting compound  $K_3NbF_8$  is formed in the system. Its melting temperature differs in the range of 760–770 °C.<sup>9-11</sup> The phase diagram of binary systems  $KF(1) + Nb_2O_5$ - $(3)^{8,13,14}$  and  $K_2NbF_7(2) + Nb_2O_5(3)^{14}$  and ternary system  $KF(1) + K_2NbF_7(2) + Nb_2O_5(3)^{14}$  have been determined as well.

In the present work, the densities of binary systems  $K_2$ -NbF<sub>7</sub>(2) + Nb<sub>2</sub>O<sub>5</sub>(3) and KF(1) + Nb<sub>2</sub>O<sub>5</sub>(3) and ternary system KF(1) + K<sub>2</sub>NbF<sub>7</sub>(2) + Nb<sub>2</sub>O<sub>5</sub>(3) have been measured. On the basis of the obtained data, we calculated the molar volumes of melts and partial molar volume of Nb<sub>2</sub>O<sub>5</sub>

in different binary systems. Results have been interpreted in terms of the interaction of components.

## **Experimental Section**

The density of the investigated melts was measured using the Archimedean method. A platinum vessel suspended on a platinum wire (in the leading tube) was used as the measuring body. After the sample melted, the measuring body was immersed in the melt, and the surface of the melt was always kept 2 mm over the top of the sphere. The depth of immersion was continuously monitored and controlled using the electrical contact. The dependence of the vessel volume on temperature was determined by calibration using molten NaCl and KF. For the measuring device control and the evaluation of experimental data, an on-line PC was used. A detailed description of the measuring device used is given elsewhere.<sup>15</sup>

For the preparation of the samples, the following chemicals were used: KF (Lachema, 99%), Nb<sub>2</sub>O<sub>5</sub> (Aldrich, 99.9%), and K<sub>2</sub>NbF<sub>7</sub> (prepared in Apatity, min. 99%). KF and K<sub>2</sub>NbF<sub>7</sub> were dried in vacuum at 180 °C for 24 h. Handling of all salts was done in a glovebox under a dry, inert atmosphere.

The measurements were carried out in the temperature interval of approximately (100 to 150) °C. The samples were heated to ~170 °C above the primary crystallization temperature. Then the first measurement in the cooling direction was performed until reaching a temperature of ~20 °C above the temperature of primary crystallization. The second measurement was done in the heating direction, and the third measurement was done again in the cooling direction. The density results were automatically registered by the measuring device every 30 s for each melt. The densities were measured for each system up to 20 mol % of Nb<sub>2</sub>O<sub>5</sub> because of the solubility of Nb<sub>2</sub>O<sub>5</sub> in the temperature range interesting for technical applications.

The temperature dependence of the density was expressed in the form of the linear equation

$$\rho/\text{g·cm}^{-3} = a - bt/^{\circ}\text{C} \tag{1}$$

where  $\rho$  is the density and *t* is the temperature. For each measurement, one heating and two cooling curves were recorded. However, in some cases the last cooling run or last cooling and heating runs were useless. The reasons are quite unclear. The evaporation of the sample seems to

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Table 1.	Regression	Coefficients a and b	and Standard D	<b>Deviations of th</b>	e Density of the	<b>Investigated Syste</b>	m KF(1) -
K <sub>2</sub> NbF <sub>7</sub> (	$(2) + Nb_2O_5(3)$	and Mean Regress	on Coefficients a	a and b and the	Standard Devia	ations Used for Eq	uation 1 <sup>a</sup>

			a		$b imes 10^4$				
$x_1$	$x_2$	$x_3$	g•cm <sup>-3</sup>	ā	g·cm <sup>-3</sup> ·°C	$ar{b} imes 10^4$	${ m SD} imes 10^4$	${ m SD} imes 10^4$	t/°C
				2.540					
0.980	0.000	0.020	2.5497	5	6.5028	6.3796	4.88	5.51	860-1000(c)
0.980	0.000	0.020	2.5397		6.3269		7.22		860–1000(h)
0.980	0.000	0.020	2.5320		6.3089		4.44		860-1000(c)
				2.633					
0.965	0.000	0.035	2.6484	6	6.6194	6.4333	1.78	3.93	850-1000(c)
0.965	0.000	0.035	2.6291		6.3379		6.96		850-1000(h)
0.965	0.000	0.035	2.6233		6.3426		3.05		850-1000(c)
0.050	0.000	0.050	0.0701	2.837	0.0550	F 4950	4.10	0.00	050 1000()
0.950	0.000	0.050	2.8731	6	8.0579	7.4258	4.18	3.86	850 - 1000(c)
0.950	0.000	0.050	2.7693		6.5906		4.05		850-1000(n)
0.950	0.000	0.050	2.8704	9.096	7.6290		3.33		850-1000(c)
0.025	0.000	0.075	2 0855	2.920	8 3904	7 5399	5.64	2 77	825 - 075(a)
0.925	0.000	0.075	2.9000	0	6.5204	1.0022	0.04	0.11	825 - 975(c)
0.925	0.000	0.075	2.0002		7 5967		2.42		825 - 975(n)
0.325	0.000	0.075	2.3412	2 976	1.5501		0.20		020 970(C)
0.900	0.000	0.100	2 9294	2.570	6 8290	7 1219	3 50	3 10	870 - 1000(c)
0.900	0.000	0.100	3 0262	4	7 4996	1.1210	3.65	0.10	870 - 1000(b)
0.900	0.000	0.100	2 9731		7 0372		2.16		870 - 1000(n)
0.000	0.000	0.100	2.0101	3.232	1.0012	7.575	2.10		010 1000(0)
0.850	0.000	0.150	3.2868	3	8.2561	8	10.10	8.73	900 - 1000(c)
0.850	0.000	0.150	3.2135		7.2916		11.20		900 - 1000(h)
0.850	0.000	0.150	3.1968		7.1796		4.88		900 - 1000(c)
				3.469					
0.800	0.000	0.200	3.6117	9	9.3346	7.8652	6.55	4.89	910-1050(c)
0.800	0.000	0.200	3.3282		6.3957		3.22		910-1050(h)
				3.223		10.647			
0.000	0.950	0.050	3.2290	8	10.7713	3	3.66	3.66	730-870(c)
0.000	0.950	0.050	3.2186		10.5232		3.66		730–870(h)
				3.221		10.860			
0.000	0.900	0.100	3.2130	5	10.8350	6	3.24	3.24	770-650(c)
0.000	0.900	0.100	3.2301		10.8862		3.24		770-650(c)
0.000	0.050	0.150	0.1500	3.175	10.0000	10.540	0.05	0.05	800 <b>5</b> 00()
0.000	0.850	0.150	3.1522	5	10.3288	2	8.95	8.95	680 - 720(c)
0.000	0.850	0.150	3.1988	9.007	10.7515	10 100	8.95		680-720(c)
0.000	0.000	0.900	2 2072	3.297	19 1907	12.120	95.90	95 20	EE0 (200(a)
0.000	0.800	0.200	3.2973	3 9.790	12.1207	1	20.00	20.00	550 - 620(c)
0.675	0 225	0 100	2 8248	2.700	5 6944	5 2003	1 97	3.03	850 - 950(c)
0.675	0.225	0.100	2.8240	1	5 5466	5.2005	4.27	5.05	890 - 930(b)
0.675	0.225	0.100	2.6998		4 3599		3.20		850 - 950(c)
0.015	0.220	0.100	2.0550	3 046	4.0000		0.20		000 000(0)
0.450	0.450	0.100	3.0622	4	7.6378	7.3821	4.96	4.61	760 - 910(c)
0.450	0.450	0.100	3.0307	-	7.1264		4.25		760 - 860(c)
				3.023					
0.225	0.675	0.100	3.0078	4	7.0145	7.2772	5.54	5.07	730-830(c)
0.225	0.675	0.100	3.0390		7.5398		4.60		730-831(c)
				3.100					
0.600	0.200	0.200	3.1000	0	5.6805	5.6805	8.01	8.01	850-960(c)
				3.186					
0.400	0.400	0.200	3.1861	1	7.1274	7.1274	7.80	7.80	770-900(c)
				3.914		17.468			
0.200	0.600	0.200	3.9149	9	17.4681	1	14.10	14.10	745-760(c)

 $^{a}$  c - cooling; h - heating.

be one of the reasons. Even a small amount of condensed vapor can result in friction between the platinum wire (on which the measuring body was suspended) and leading tube. For each errorless curve, coefficients a and b were obtained. The values of constants a and b together with the standard deviations of approximations, obtained by the linear regression analysis of the experimentally obtained data, are given in Table 1. The results of measurements at 850 °C, 900 °C, and 950 °C are summarized in Table 2.

## **Results and Discussion**

**System KF(1)** +  $Nb_2O_5(3)$ . The density of the system KF(1) +  $Nb_2O_5(3)$  is shown in Figure 1. The density increases with increasing content of  $Nb_2O_5$ . All three values of the density were averaged for each composition. The

concentration dependence of the molar volume of this system at 900  $^{\circ}\mathrm{C}$  can be described by the equation

$$V/\text{cm}^3 \cdot \text{mol}^{-1} = (30.95 + 27.15x_3)$$
  
SD = 0.13 cm<sup>3</sup>·mol<sup>-1</sup> (2)

By differentiating eq 2 according to  $x_3$  and introducing the result into the equation

$$\bar{V}_3 = V + x_1 \left(\frac{\partial V}{\partial x_3}\right) \tag{3}$$

we can obtain the partial molar volume of  $\mathrm{Nb}_2\mathrm{O}_5$  in the form

$$\bar{V}_3$$
/cm<sup>3</sup>·mol<sup>-1</sup> = 58.08 ± 0.10 (4)



**Figure 1.** Density,  $\rho$ , of the system KF(1) + Nb<sub>2</sub>O<sub>5</sub>(3).  $\Box$ , 850 °C;  $\bigcirc$ , 900 °C;  $\triangle$ , 950 °C.

Table 2. Calculated Data for the Density of the Molten System KF(1) +  $K_2NbF_7(2)$  +  $Nb_2O_5(3)$  at 850 °C, 900 °C, and 950 °C

			$ ho/ m g\cdot cm^{-3}$		
$x_1$	$x_2$	$x_3$	850 °C	900 °C	950 °C
0.980	0.000	0.020	1.9969	1.9663	1.9344
0.965	0.000	0.035	2.0857	2.0546	2.0224
0.950	0.000	0.050	2.1882	2.1693	2.1321
0.925	0.000	0.075	2.2783	2.2487	2.2111
0.900	0.000	0.100	2.3490	2.3352	2.2996
0.850	0.000	0.150	2.5850	2.5505	2.5126
0.800	0.000	0.200	2.8182	2.7620	2.7227
0.000	0.950	0.050	2.3135	2.2656	2.2123
0.000	0.900	0.100	2.2920	2.2441	2.1898
0.000	0.850	0.150	2.2743	2.2269	2.1742
0.000	0.800	0.200	2.2671	2.2671	2.2671
0.675	0.225	0.100	2.3408	2.3121	2.2861
0.450	0.450	0.100	2.4130	2.3820	2.3451
0.225	0.675	0.100	2.4116	2.3685	2.3321
0.600	0.200	0.200	2.6172	2.6172	2.6172
0.400	0.400	0.200	2.5802	2.5802	2.5802
0.200	0.600	0.200	2.4301	2.4301	2.4301

As can be seen, no dependence on KF was obtained, and the partial molar volume of  $Nb_2O_5$  represents the molar volume of  $Nb_2O_5$ , which means that the volume properties of the above binary behave ideally in the composition range up to 20 mol %  $Nb_2O_5$ . However, it does not mean that any reactions take place in this system.

**System**  $K_2NbF_7(2) + Nb_2O_5(3)$ . The density of the system  $K_2NbF_7(2) + Nb_2O_5(3)$  is shown in Figure 2. The molar volumes have been calculated from the first cooling values of the densities because after the second heating of the sample slight evaporation was detected. The density of the system  $K_2NbF_7(2) + Nb_2O_5(3)$  decreases with increasing content of  $Nb_2O_5$ . The concentration dependence of the molar volume at 900 °C can be described by the equation

$$V/\text{cm}^3 \cdot \text{mol}^{-1} = (132.56 + 23.88x_3 - 76.75x_3^2)$$
  
SD = 0.20 cm<sup>3</sup>·mol<sup>-1</sup> (5)



**Figure 2.** Density,  $\rho$ , of the system K<sub>2</sub>NbF<sub>7</sub>(2) + Nb<sub>2</sub>O<sub>5</sub>(3).  $\Box$ , 850 °C;  $\bigcirc$ , 900 °C;  $\triangle$ , 950 °C.



**Figure 3.** Density,  $\rho$ , of the system KF(1) + K<sub>2</sub>NbF<sub>7</sub>(2) + Nb<sub>2</sub>O<sub>5</sub>-(3) at 900 °C. Values are in g·cm<sup>-3</sup>.

The partial molar volume of  $Nb_2O_5\ can be described by the equation$ 

$$\bar{V}_{3}$$
/cm<sup>3</sup>·mol<sup>-1</sup> = (79.68 + 76.75x<sub>2</sub><sup>2</sup>) (6)

and for  $x_2 \rightarrow 1$  the partial molar volume of Nb<sub>2</sub>O<sub>5</sub> has the value of  $\bar{V}_3 = (156.43 \pm 0.22) \text{ cm}^3 \cdot \text{mol}^{-1}$ . In this binary system, the value of the partial molar volume of Nb<sub>2</sub>O<sub>5</sub> is considerably higher than in the previous system. Moreover, the ideal behavior of the volume properties can be excluded here, probably because of the formation of more voluminous species.

System  $KF(1) + K_2NbF_7(2) + Nb_2O_5(3)$ . The density of the system  $KF(1) + K_2NbF_7(2) + Nb_2O_5(3)$  is shown in Figure 3. The density in the system increases from KF through  $K_2NbF_7$  to  $Nb_2O_5$ . Generally, the dependence of the molar volume of the ternary system on composition at constant temperature can be described by the Redlich– Kister-type equation

$$V = \sum_{i=1}^{3} x_i V_i + \sum_{i \neq j}^{3} x_i x_j \sum_{n=0}^{k} A_{nij} x_j^n + B x_1^p x_2^q x_3^r$$
(7)

where p, q, r, and n are adjustable integers. The first term represents ideal behavior, the second one describes the binary interactions, and the third one represents the interaction of all three components.

In our case, constants  $V_i$ ,  $A_{nij}$ , and B in eq 7 were calculated using multiple linear regression analysis omit-



**Figure 4.** Molar volume,  $V_m$ , of  $KF(1) + K_2NbF_7(2) + Nb_2O_5(3)$  at 900 °C. Values are in cm<sup>3</sup>·mol<sup>-1</sup>.

Table 3. Calculated Coefficients  $V_i$  and  $A_{nij}$  and the Standard Deviations of the Density of the System KF(1) + K<sub>2</sub>NbF<sub>7</sub>(2) + Nb<sub>2</sub>O<sub>5</sub>(3) at Different Temperatures

coefficient		t/°C	
g•mol <sup>-3</sup>	850	900	950
$V_1$	$30.69 \pm 0.34$	$31.25\pm0.38$	$31.83 \pm 0.43$
$V_2$	$131.73\pm1.69$	$134.80\pm1.88$	$138.01\pm2.16$
$V_3$	$110.30\pm2.12$	$116.17\pm2.43$	$120.79\pm2.84$
$A_{112}$	$-47.73\pm9.28$	$-53.64\pm10.28$	$-60.06 \pm 11.82$
$A_{023}$	$8.30\pm2.10$	$6.75 \pm 2.31$	$6.93 \pm 2.58$
$A_{013}$	$-65.20\pm4.67$	$-72.33\pm5.21$	$-77.86\pm6.00$
SD	0.93	1.03	1.18

ting the statistically nonimportant terms on the 0.99 confidence level. The following final equation was obtained:

$$V/\text{cm}^3 \cdot \text{mol}^{-1} = x_1 V_1 + x_2 V_2 + x_3 V_3 + A_{112} x_1 x_2^2 + A_{023} x_2 x_3 + A_{013} x_1 x_3$$
(8)

The values of constants  $V_i$  and  $A_{nij}$  in eq 8 as well as the standard deviations of approximation for temperatures 850 °C, 900 °C, and 950 °C are given in Table 3.

The molar volume of the system  $KF(1) + K_2NbF_7(2) + Nb_2O_5(3)$  at the temperature of 900 °C is shown in Figure 4.

From the parameters of eq 8, the molar volumes of the system (1/1) KF/K<sub>2</sub>NbF<sub>7</sub> + Nb<sub>2</sub>O<sub>5</sub> were calculated at 900 °C up to 20 mol % Nb<sub>2</sub>O<sub>5</sub> with a step of 1 mol %. Let us define the system (1/1) KF/K<sub>2</sub>NbF<sub>7</sub> to be K<sub>3</sub>NbF<sub>8</sub>(4) and its molar fraction to be  $x_4$ . Then the equation representing the dependence of the molar volume on the composition of Nb<sub>2</sub>O<sub>5</sub> in the system K<sub>3</sub>NbF<sub>8</sub>(4) + Nb<sub>2</sub>O<sub>5</sub>(3) was calculated:

$$V/cm^{3} \cdot mol^{-1} = (74.48 + 22.16x_{3} + 14.69x_{3}^{2})$$
  
SD/cm<sup>3</sup> · mol^{-1} = 1.25 × 10^{-3} (9)

Then the partial molar volume of  $Nb_2O_5$  can be described by the equation

$$\bar{V}_{3}$$
/cm<sup>3</sup>·mol<sup>-1</sup> = (111.33 - 14.69 $x_{4}^{2}$ ) (10)

and for  $x_4 \rightarrow 1$  the partial molar volume of Nb<sub>2</sub>O<sub>5</sub> has the value of  $V_3 = 96.64 \text{ cm}^3 \cdot \text{mol}^{-1}$ . In this binary system, the value of the partial molar volume of Nb<sub>2</sub>O<sub>5</sub> is considerably higher than in the first one but lower as in the second one. It can be said that the partial molar volume increase in the series KF(1) + Nb<sub>2</sub>O<sub>5</sub>(3) < K<sub>3</sub>NbF<sub>8</sub>(4) + Nb<sub>2</sub>O<sub>5</sub>(3) < K<sub>2</sub>NbF<sub>7</sub>(2) + Nb<sub>2</sub>O<sub>5</sub>(3).

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